

# The Linearized Augmented Planewave (LAPW) Method

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$$E_T[\rho] = T_s[\rho] + E_{ei}[\rho] + E_H[\rho] + E_{xc}[\rho] + E_{ii}$$

$$\{T_s + V_{ks}[\rho, r]\} \varphi_I(r) = \varepsilon_i \varphi_i(r)$$

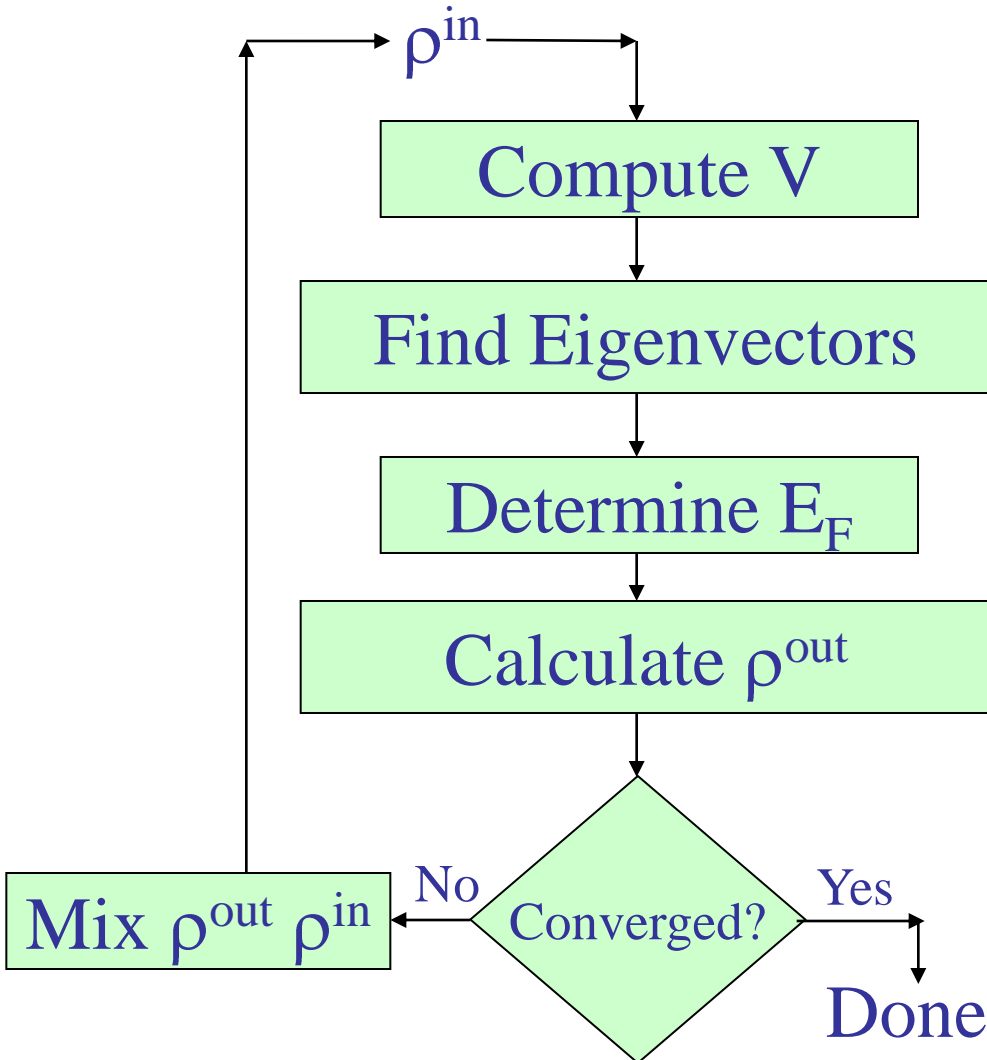
***Need tools that are reliable and predictive.***



# DFT ALGORITHMS

- Find  $\varphi_I$  and  $\rho$  to solve:

$$\{ T_s + V_{KS}[\rho, r] \} \varphi_I(r) = \varepsilon_i \varphi_i(r)$$



## Standard Solution:

- Expand  $\varphi_i$  in a basis  $\{\phi_j\}$ .
- Many methods, PW, FE, LAPW, LMTO, LCAO ...
- For fixed  $V_{KS}$  get a linear algebra problem.

(eigenvalue).

$$\langle \phi | H | \phi \rangle x_i = \varepsilon_i \langle \phi | \phi \rangle x_i$$

- Iterate to find self-consistent  $\rho$ .

## Some Numbers:

- #  $\varphi_i \sim 10$  / atom.
- #  $\phi_j \sim 10$ 's - 1000's / atom.
- # atoms (State of the Art): 100 – 1000's.

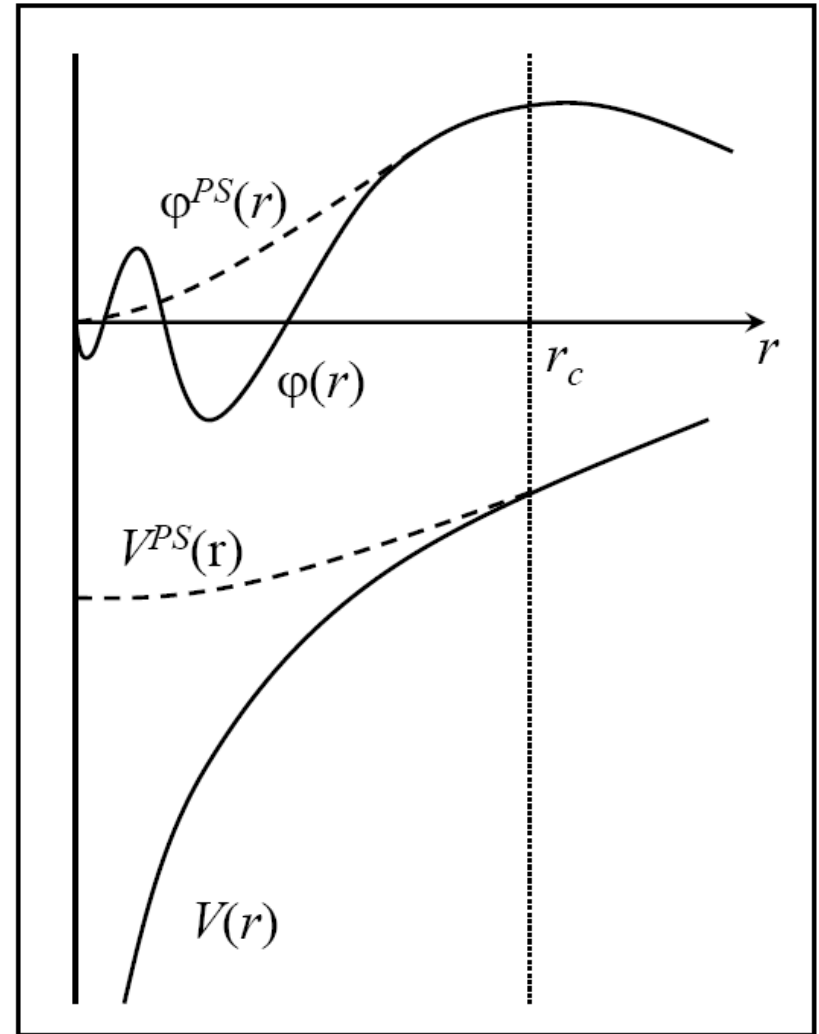
# Motivation for Augmentation

Schrödinger Equation:

$$(T+V-\varepsilon)\varphi = 0$$

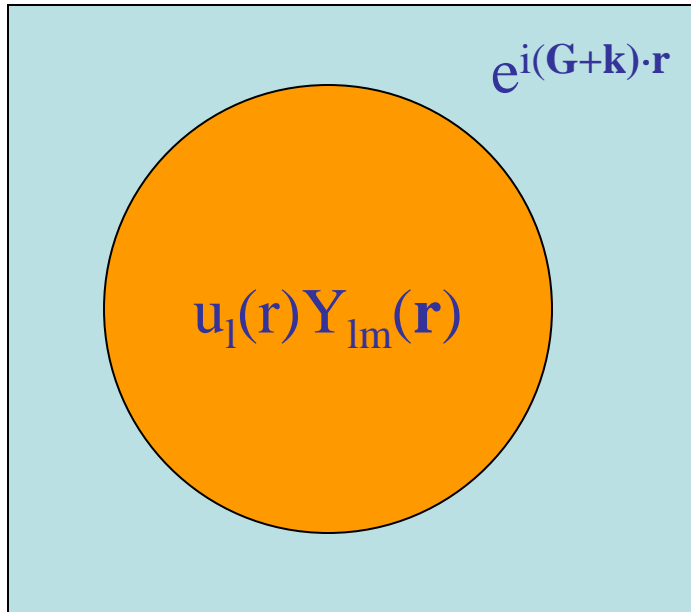
For valence states:  $\varepsilon$  is small  $\rightarrow$

*$T\varphi$  is also small except where  $V$  is strong, i.e. near the nucleus.*



# Augmented Planewave (APW) Method

• J.C. Slater, *Phys. Rev.* **51**, 846 (1937); *Phys. Rev.* **81**, 385 (1951).



Divide Space Into 2 Regions:

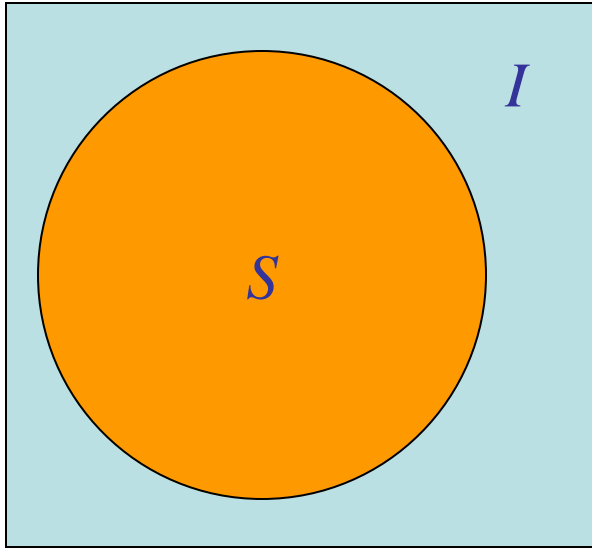
- Atom Centered Spheres
- Interstitial

“Basis” Consists of Planewaves in the Interstitial and Radial Functions in the Spheres.

$$\varphi(\mathbf{r}) = \begin{cases} \Omega^{-1/2} \sum_{\mathbf{G}} c_{\mathbf{G}} e^{i(\mathbf{G}+\mathbf{k})\cdot\mathbf{r}} & \mathbf{r} \in \text{Interstitial (I)} \\ \sum_{lm} A_{lm} u_l(r) Y_{lm}(\mathbf{r}) & \mathbf{r} \in \text{Sphere (S)} \end{cases}$$

- $u_l(r)$  are the radial solutions of Schrodinger’s equation at the energy of interest (i.e. the band energy).

# Efficiency & Accuracy



(1) Very efficient basis set.

(2) Represent all quantities as generally as possible in all regions of space and make only controlled approximations.

## Spheres: Atomic-like treatment

- Numerical radial functions times  $Y_{lm}$ : can increase  $l_{\max}$
- Angular integrals are fast due to orthogonality of the  $Y_{lm}$

## Interstitial: Free space like treatment

- Planewave expansions.
- Integrals are fast due to FFT properties
  - Step function (cut out spheres) can be done exactly up to finite  $G_{\max}$  by convolution with pre-computed  $U(G)$

# Augmented Planewave (APW) Method

$$\varphi(\mathbf{r}) = \begin{cases} \Omega^{-1/2} \sum_{\mathbf{G}} c_{\mathbf{G}} e^{i(\mathbf{G}+\mathbf{k})\cdot\mathbf{r}} & \mathbf{r} \in \text{Interstitial (I)} \\ \sum_{lm} A_{lm} u_l(r) Y_{lm}(\mathbf{r}) & \mathbf{r} \in \text{Sphere (S)} \end{cases}$$

## Key points:

1. The  $A_{lm}$  are not variational parameters. They are determined by a matching condition. That is the value of the basis functions,  $\phi_{\mathbf{k}+\mathbf{G}}$  is fixed to be continuous at the sphere boundary.
2. The full crystal potential can be used because one may show that the  $u_l$  are orthogonal to “core” states.

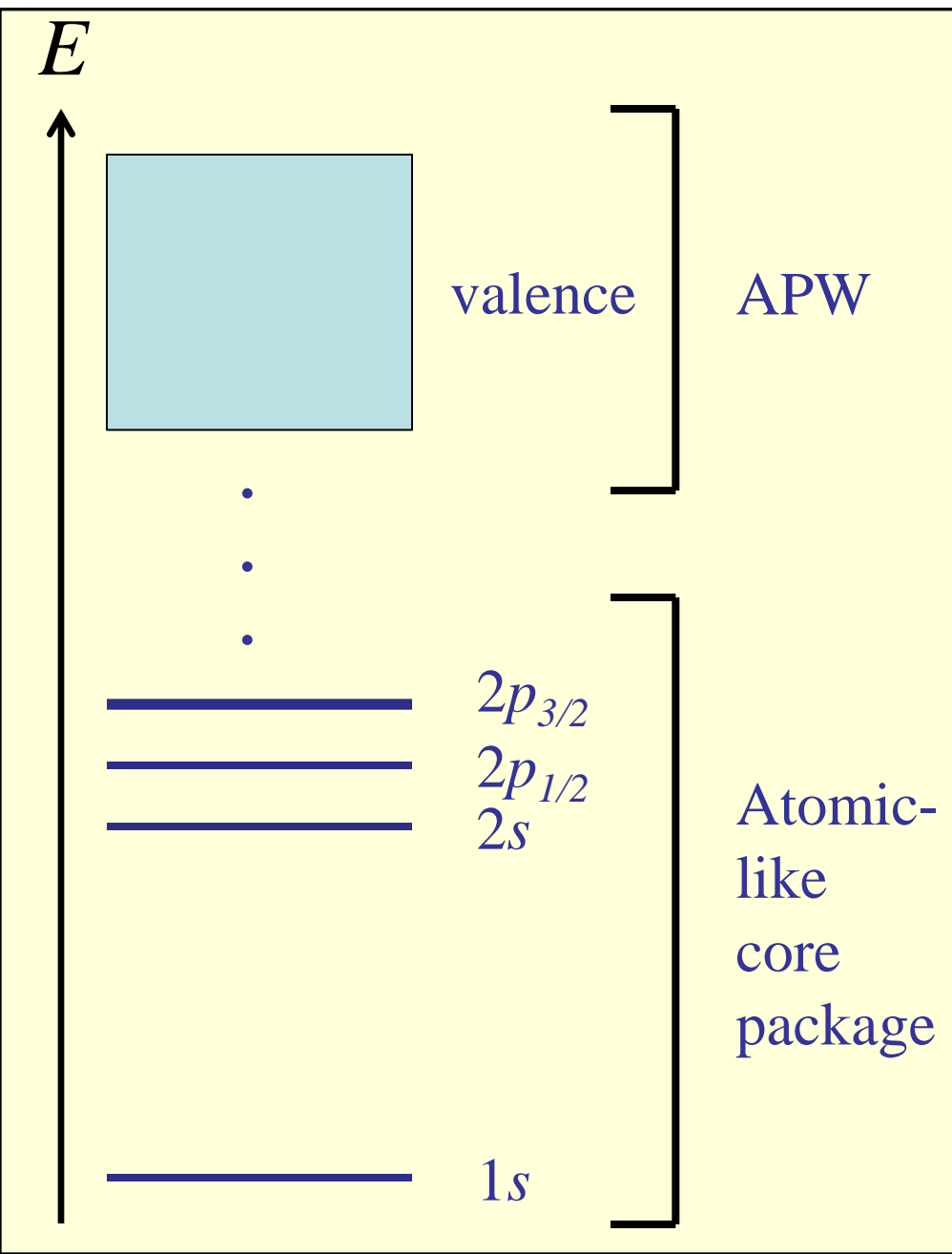
$$[ -d^2/dr^2 + l(l+1)/r^2 + V(r) - E_l ] ru_l(r) = 0$$

So:

$$(E_2 - E_1) r u_1 u_2 = u_2 (d^2 ru_1/dr^2) - u_1 (d^2 ru_2/dr^2)$$

Integrate by parts to get overlap of  $u_1$  and  $u_2$ . They are orthogonal if one of them is 0 on the sphere boundary.

# APW: An All-Electron Method



The  $u_l(\mathbf{r}) Y_{lm}(\mathbf{r})$  are orthogonal core states.

→ Can use this basis to obtain true valence states in the real potential.

- (1) Calculate core states separately in each SCF cycle.
- (2) Use the same potential for core and valence and calculate the charge density from the sum of these.

# Augmented Planewave (APW) Method

## Another Interesting Point:

- Since the basis functions are indexed by  $\mathbf{k}+\mathbf{G}$  one imagines a connection with planewave pseudopotential formalisms.

$$\langle A\phi | H | A\phi \rangle_{\mathbf{x}} = \varepsilon \langle A\phi | A\phi \rangle_{\mathbf{x}} \quad \equiv \quad \langle \phi | \underbrace{A^\dagger H A}_{H^{\text{PS}}} | \phi \rangle_{\mathbf{x}} = \varepsilon \langle \phi | \underbrace{A^\dagger A}_{S^{\text{PS}}} | \phi \rangle_{\mathbf{x}}$$

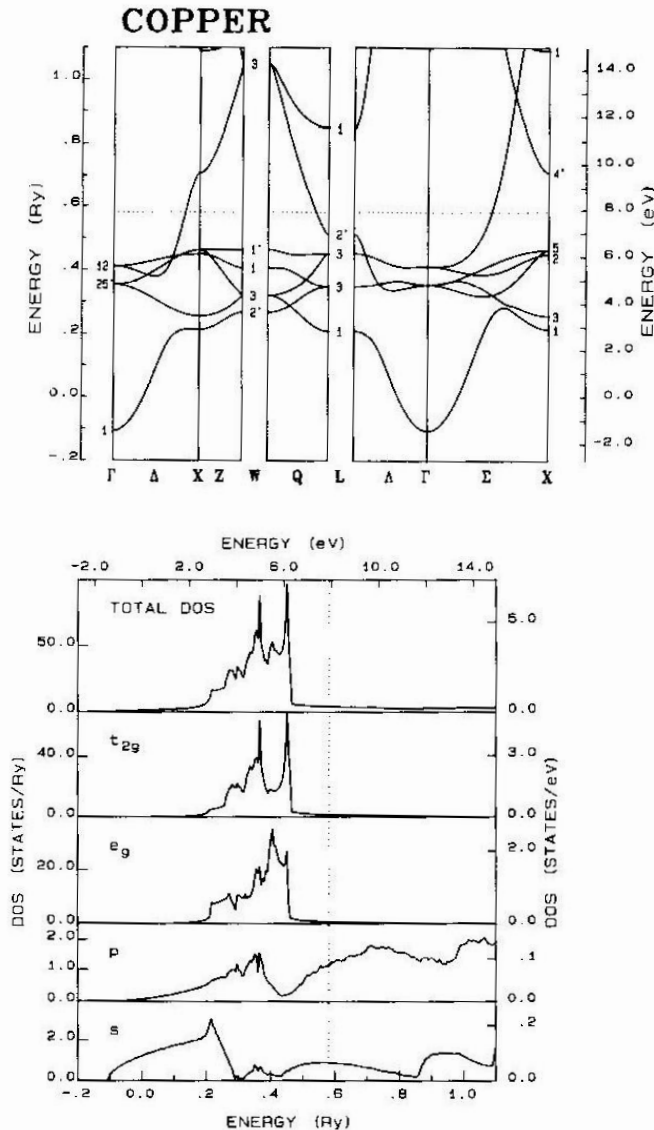
- So this is like non-norm-conserving pseudopotential.
- However, it is highly non-transferable:
  - Cannot be used at another energy (because  $u$  is very energy dependent -  $\partial u / \partial E$  is usually large).
  - Cannot be used for a different potential.
- **Result:** *The APW method as written requires use of an energy dependent secular equation and is not practical for more than simple solids.*



# The APW Method as a Pseudopotential

- It's highly non-transferable, but it is soft!

← APW Band Structure of Cu using a planewave cutoff of 8.4 Ry.



*There is a trade-off between transferability and softness (nothing is free). The story of linearization and local orbitals is related to this.*

# Problems with the APW Method

- 1) Must solve secular equation for each energy band: prohibitive for many bands. No clear way to make full-potential.
- 2) Asymptote problem: cannot match at energies where  $u(r)$  is zero on the sphere boundary. This will in general happen at some energy – particular problem for  $d$  and  $f$  band materials.

# The Linearized Augmented Planewave (LAPW) Method

O.K. Andersen, Phys. Rev. B **12**, 3060 (1975).

## Key Ideas:

- The problem with the APW method is the energy dependence of the secular equation which is a result of the energy dependence of the augmenting function.
- Solution: Add variational freedom: particularly  $\dot{u}(r) = \partial u(r)/\partial E$ .

$$\varphi(\mathbf{r}) = \begin{cases} \Omega^{-1/2} \sum_{\mathbf{G}} c_{\mathbf{G}} e^{i(\mathbf{G}+\mathbf{k})\cdot\mathbf{r}} & \mathbf{r} \in \text{I} \\ \sum_{lm} (A_{lm} u_l(r) + B_{lm} \dot{u}_l(r)) Y_{lm}(\mathbf{r}) & \mathbf{r} \in \text{S} \end{cases}$$

- Where  $A_{lm}$  *and*  $B_{lm}$  are determined by matching the value and derivative of the basis functions at the sphere boundary.

# THE LAPW METHOD

Results of adding  $\hat{u}_l$  to the basis:

1. Basis is flexible enough to use a single diagonalization (energy errors are now  $O(\delta^4)$ ).
2. Must have additional matching conditions to connect both  $u$  and  $\hat{u}$  to the planewaves. This means that for a given level of convergence, more planewaves are needed.
3. The transferability also extends to variations in the potential: this enables full-potential methods.

*The full potential, all electron, nature combined with the flexible basis (fully flexible in the interstitial) made the (F)LAPW method the state of the art for calculating electronic structures, especially for transition elements and their compounds – Many groups developed codes 1980 – present.*

# Early Impact

Many works starting in 1980's showing predictive calculations for complex materials and surfaces with  $d$  and  $f$  elements.

VOLUME 57, NUMBER 26

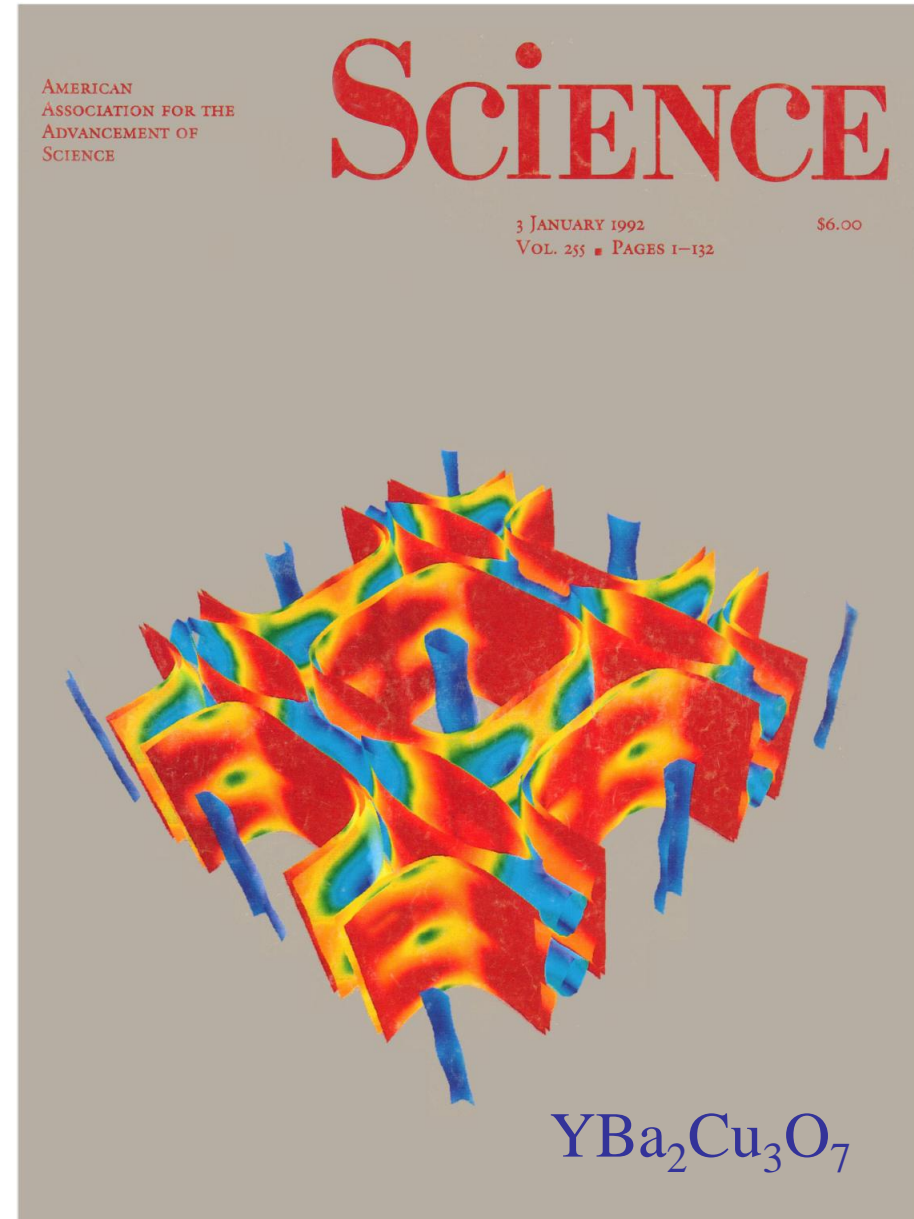
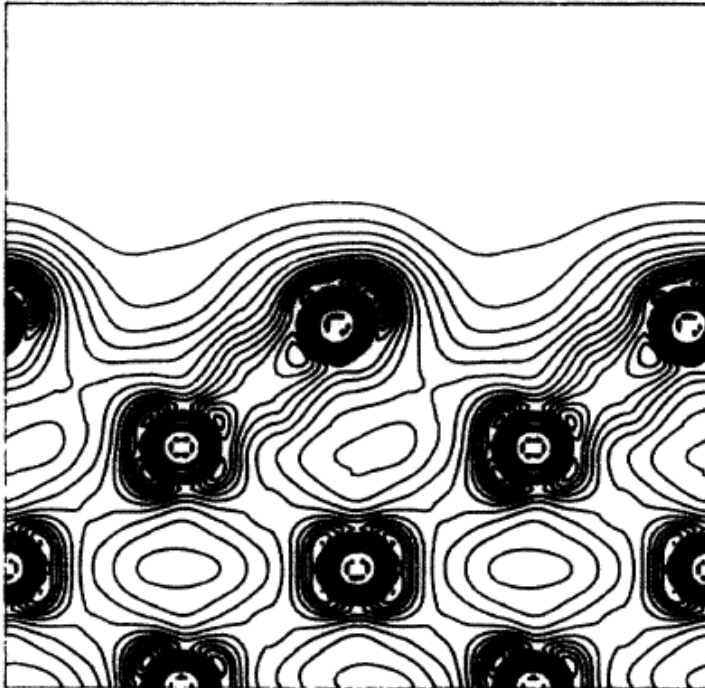
PHYSICAL REVIEW LETTERS

29 DECEMBER 1986

## Instability of the Ideal Tungsten (001) Surface

David Singh, Su-Huai Wei,<sup>(a)</sup> and Henry Krakauer

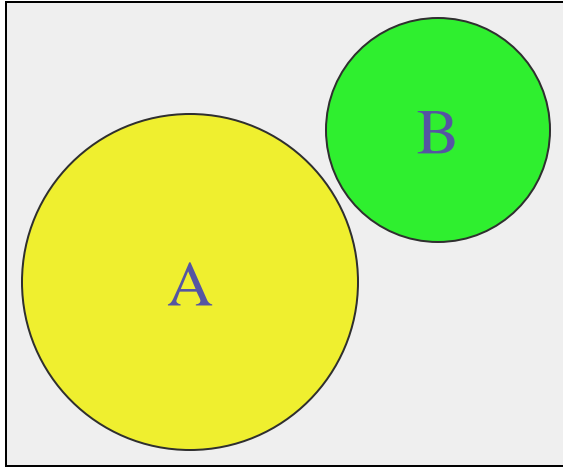
*Department of Physics, College of William and Mary, Williamsburg, Virginia 23185*



# PROPERTIES OF THE LAPW METHOD

- All electron method: Core states are included.
  - $\varphi$  is the true wavefunction,  $\rho$  is the true charge density ...
  - Can calculate properties that depend on the details of the wavefunction near the nucleus: EFG's *etc.*
  - Relativity can be included – scalar relativistic, spin-orbit ...
  - No special treatment for core-valence interactions is needed.
- Atom centered representation:
  - LDA+U, interpretation of transition element orbital populations.
  - Matrix elements are complicated.
  - IBS terms in forces, linear response ...
  - Basis functions are extended – not very amenable to  $O(N)$  ...

# CHOICE OF SPHERE RADII



Size of basis,

$$n_b \propto G_{\max}^3$$

Compute time,

$$t \propto n_b^3 \propto G_{\max}^9$$

For most atoms, with “normal radii”, a given level of convergence is reached for a certain, atom dependent value of  $rG_{\max}$ .

Typical  $rG_{\max}$  values for good convergence (always check):

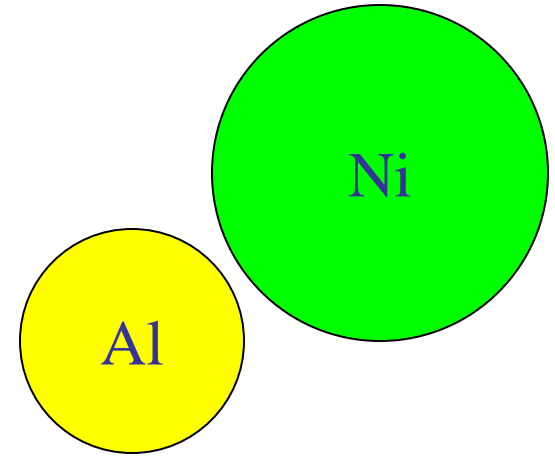
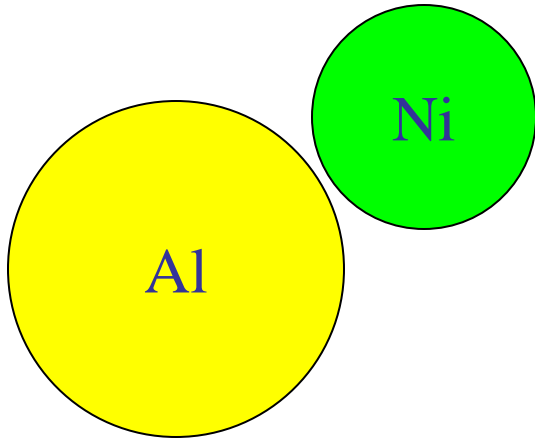
Transition elements:	9	<i>Should consider</i>
f-electron materials:	9.5	<i>in setting radii, which</i>
Simple elements (B,C,N,O)	7	<i>are computational</i>
Simple metals (Al, Si, ...)	6	<i>not physical parameters.</i>

# Example (B2 NiAl)

Chemical Sense

vs.

Computational Sense



$$r_{Al} = 2.8 \text{ bohr}$$

$$r_{Ni} = 1.9 \text{ bohr}$$

$$r_{Al} G_{max} = 6 \rightarrow G_{max} = 2.15$$

$$r_{Ni} G_{max} = 9 \rightarrow G_{max} = 4.74$$

$$r_{Al} = 1.9 \text{ bohr}$$

$$r_{Ni} = 2.8 \text{ bohr}$$

$$r_{Al} G_{max} = 6 \rightarrow G_{max} = 3.15$$

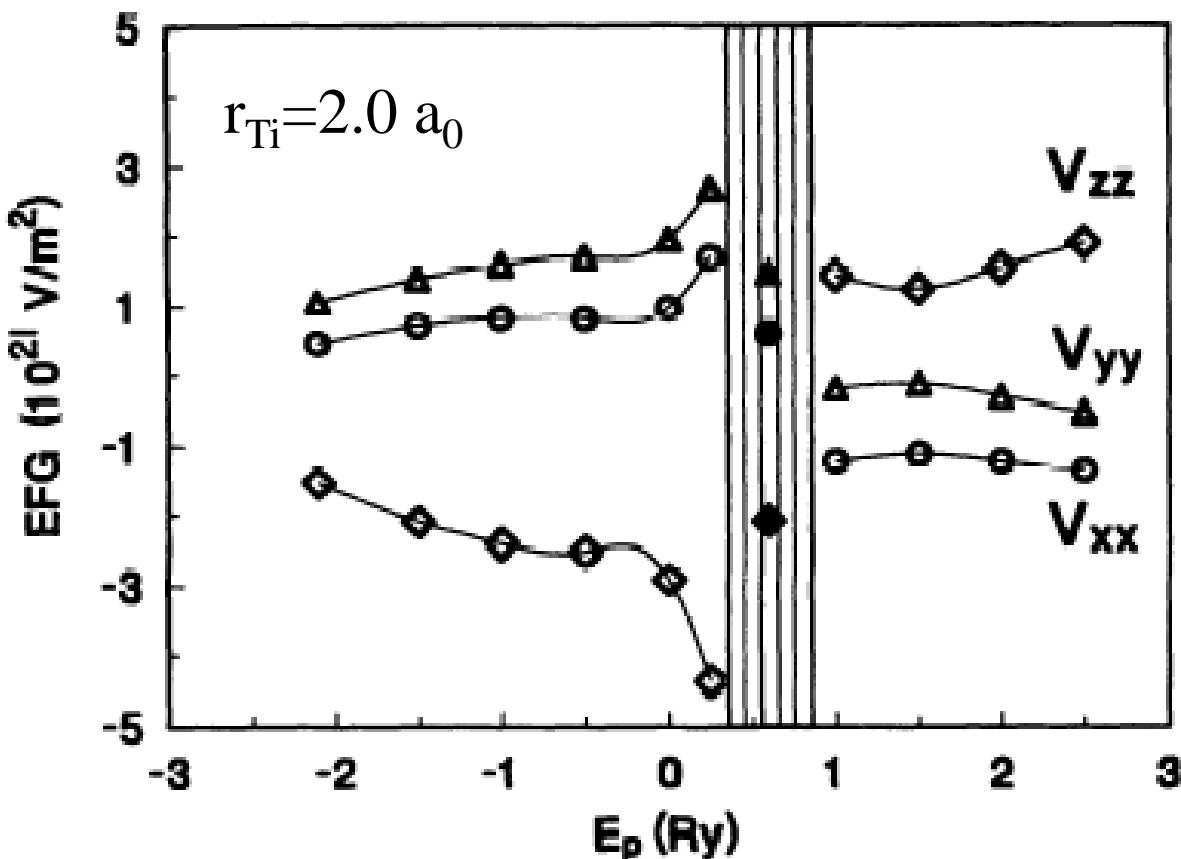
$$r_{Ni} G_{max} = 9 \rightarrow G_{max} = 3.21$$

$$(4.74/3.21)^9 = 33$$



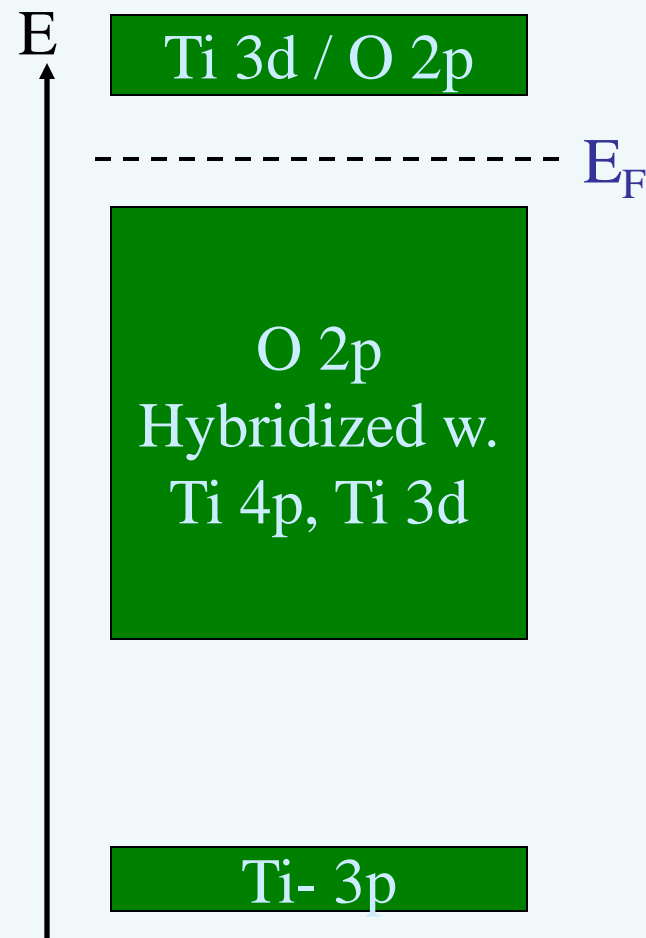
# Complications in the LAPW Method

EFG Calculation for Rutile  $\text{TiO}_2$  as a function of the Ti  $p$  linearization energy



P. Blaha, D.J. Singh, P.I. Sorantin and K. Schwarz, Phys. Rev. B **46**, 1321 (1992).

## Electronic Structure



# Complications in the LAPW Method

What went wrong?

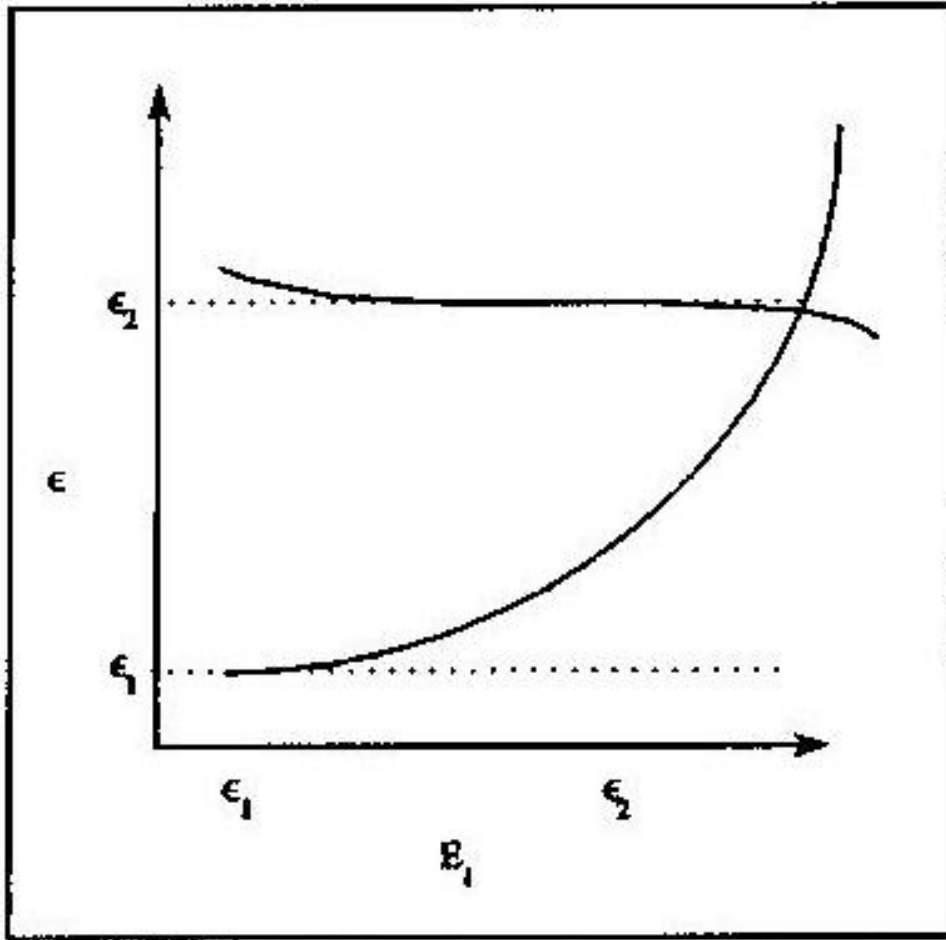


Figure 5.14 Variation of a semi-core and a valence band with  $E_l$ . The dotted lines at  $\epsilon_1$  and  $\epsilon_2$  denote the true locations of the bands.

The LAPW method requires non-overlapping spheres

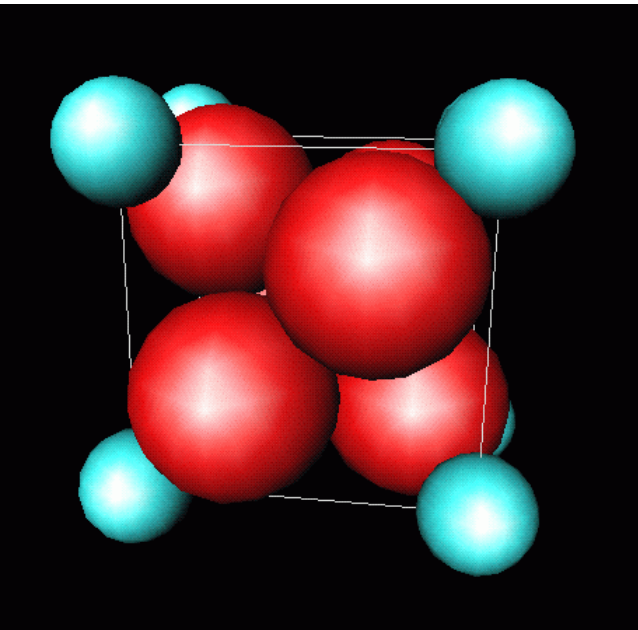
$\Rightarrow$  *There are serious limits to how large  $R_{MT}$  can be especially in oxides, nitrides, carbides.*

But for many elements there are extended core states that are not close enough to zero on the sphere boundary to have the  $u$  and  $\dot{u}$  orthogonal to them. On the other hand, the valence states may have significant contributions from the same  $l$ .

# Complications in the LAPW Method

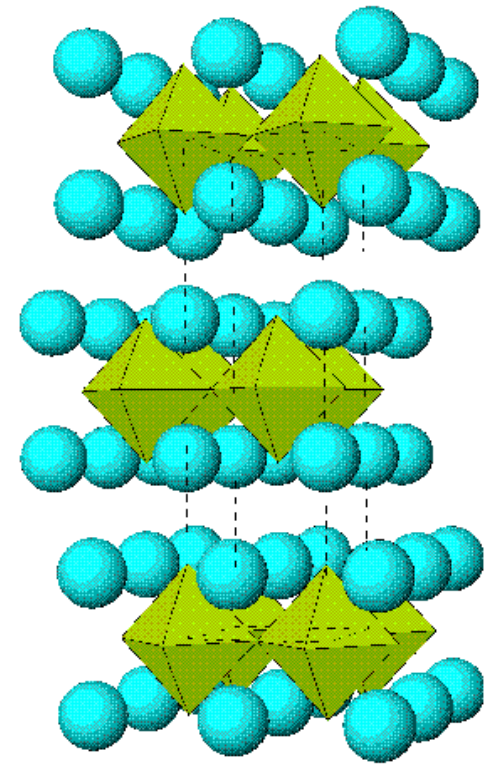
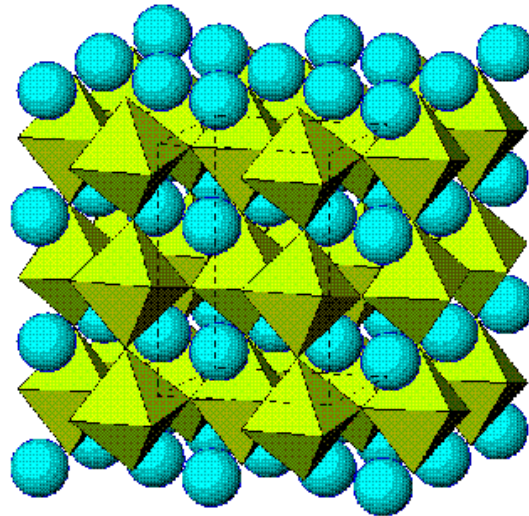
*Solution?: Use large spheres to get orthogonality to core states:*

*Unfortunately, crystal structures don't generally allow this.*



Rutile Structure

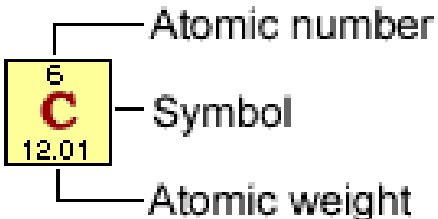
Perovskite



Layered Perovskite

# Complications in the LAPW Method

1	1 <b>H</b> 1.008	2											13	14	15	16	17	18 2 <b>He</b> 4.003													
2	3 <b>Li</b> 6.941	4 <b>Be</b> 9.012											5 <b>B</b> 10.81	6 <b>C</b> 12.01	7 <b>N</b> 14.01	8 <b>O</b> 16.00	9 <b>F</b> 19.00	10 <b>Ne</b> 20.18													
3	11 <b>Na</b> 22.99	12 <b>Mg</b> 24.31	3	4	5	6	7	8	9	10	11	12	13 <b>Al</b> 26.98	14 <b>Si</b> 28.09	15 <b>P</b> 30.97	16 <b>S</b> 32.07	17 <b>Cl</b> 35.45	18 <b>Ar</b> 39.95													
4	19 <b>K</b> 39.10	20 <b>Ca</b> 40.08	21 <b>Sc</b> 44.96	22 <b>Ti</b> 47.88	23 <b>V</b> 50.94	24 <b>Cr</b> 52.00	25 <b>Mn</b> 54.94	26 <b>Fe</b> 55.85	27 <b>Co</b> 58.93	28 <b>Ni</b> 58.69	29 <b>Cu</b> 63.55	30 <b>Zn</b> 65.39	31 <b>Ga</b> 69.72	32 <b>Ge</b> 72.61	33 <b>As</b> 74.92	34 <b>Se</b> 78.96	35 <b>Br</b> 79.90	36 <b>Kr</b> 83.80													
5	37 <b>Rb</b> 85.47	38 <b>Sr</b> 87.62	39 <b>Y</b> 88.91	40 <b>Zr</b> 91.22	41 <b>Nb</b> 92.91	42 <b>Mo</b> 95.94	43 <b>Tc</b> 98.91	44 <b>Ru</b> 101.1	45 <b>Rh</b> 102.9	46 <b>Pd</b> 106.4	47 <b>Ag</b> 107.9	48 <b>Cd</b> 112.4	49 <b>In</b> 114.8	50 <b>Sn</b> 118.7	51 <b>Sb</b> 121.8	52 <b>Te</b> 127.6	53 <b>I</b> 126.9	54 <b>Xe</b> 131.3													
6	55 <b>Cs</b> 132.9	56 <b>Ba</b> 137.3	71 <b>Lu</b> 175.0	72 <b>Hf</b> 178.5	73 <b>Ta</b> 180.9	74 <b>W</b> 183.8	75 <b>Re</b> 186.2	76 <b>Os</b> 190.2	77 <b>Ir</b> 192.2	78 <b>Pt</b> 195.1	79 <b>Au</b> 197.0	80 <b>Hg</b> 200.6	81 <b>Tl</b> 204.4	82 <b>Pb</b> 207.2	83 <b>Bi</b> 209.0	84 <b>Po</b> 209.0	85 <b>At</b> 210.0	86 <b>Rn</b> 222.0													
7	87 <b>Fr</b> 223.0	88 <b>Ra</b> 226.0	103 <b>Lr</b> 262.1	104 <b>Rf</b> 261.1	105 <b>Db</b> 262.1	106 <b>Sg</b> 263.1	107 <b>Bh</b> 264.1	108 <b>Hs</b> 265.1	109 <b>Mt</b> 268	110 <b>Uun</b> 269	111 <b>Uuu</b> 272	112 <b>Uub</b> 277	113 <b>Uut</b>	114 <b>Uuq</b> 289	115 <b>Uup</b>	116 <b>Uuh</b> 289	117 <b>Uus</b>	118 <b>Uuo</b> 293													
8																		57 <b>La</b> 138.9	58 <b>Ce</b> 140.1	59 <b>Pr</b> 140.9	60 <b>Nd</b> 144.2	61 <b>Pm</b> 146.9	62 <b>Sm</b> 150.4	63 <b>Eu</b> 152.0	64 <b>Gd</b> 157.3	65 <b>Tb</b> 158.9	66 <b>Dy</b> 162.5	67 <b>Ho</b> 164.9	68 <b>Er</b> 167.3	69 <b>Tm</b> 168.9	70 <b>Yb</b> 173.0
9																		89 <b>Ac</b> 227.0	90 <b>Th</b> 232.0	91 <b>Pa</b> 231.0	92 <b>U</b> 238.0	93 <b>Np</b> 237.0	94 <b>Pu</b> 244.1	95 <b>Am</b> 243.1	96 <b>Cm</b> 247.1	97 <b>Bk</b> 247.1	98 <b>Cf</b> 251.1	99 <b>Es</b> 252.0	100 <b>Fm</b> 257.1	101 <b>Md</b> 258.1	102 <b>No</b> 259.1



6

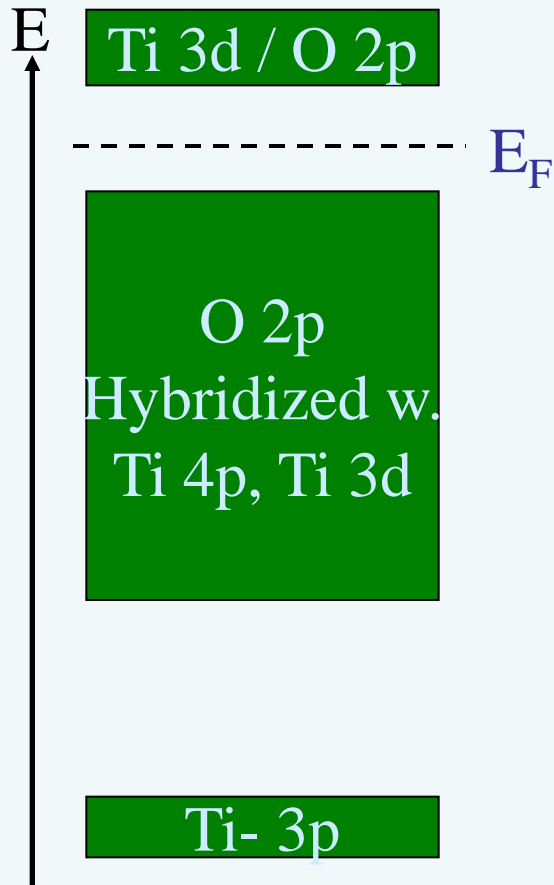
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(c) 1998  
KremerPaul

Problems with semi-core states

# ONE SOLUTION

## Electronic Structure



Treat all the states in a single energy window:

- Automatically orthogonal.
- Need to add variational freedom.
- Could invent quadratic or cubic APW methods.

$$\varphi(\mathbf{r}) = \begin{cases} \Omega^{-1/2} \sum_{\mathbf{G}} c_{\mathbf{G}} e^{i(\mathbf{G}+\mathbf{k})\cdot\mathbf{r}} \\ \sum_{lm} (A_{lm}u_l(\mathbf{r})+B_{lm}\dot{u}_l(\mathbf{r})+C_{lm}\ddot{u}_l(\mathbf{r})) Y_{lm}(\mathbf{r}) \end{cases}$$

**Problem:** This requires an extra matching condition, e.g. second derivatives continuous  $\Rightarrow$  method will be impractical due to the high planewave cut-off needed.

# THE LAPW+LO METHOD

LAPW+LO basis is:

$$\varphi(\mathbf{r}) = \left\{ \begin{array}{l} \Omega^{-1/2} \sum_{\mathbf{G}} c_{\mathbf{G}} e^{i(\mathbf{G}+\mathbf{k})\cdot\mathbf{r}} \\ \sum_{lm} (A_{lm}u_l(\mathbf{r})+B_{lm}\dot{u}_l(\mathbf{r})) Y_{lm}(\mathbf{r}) + \\ \sum_{lm} c_{lm} (A'_{lm}u_l(\mathbf{r})+B'_{lm}\dot{u}_l(\mathbf{r})+u^{(2)}_l(\mathbf{r})) Y_{lm}(\mathbf{r}) \end{array} \right.$$

The variational coefficients are: (1)  $c_{\mathbf{G}}$  and (2)  $c_{lm}$

Subsidiary (non-variational) coefficients are  $A_{lm}$   $B_{lm}$   $A'_{lm}$  &  $B'_{lm}$

- $A_{lm}$  and  $B_{lm}$  are determined by matching the value and derivative on the sphere boundary to the planewaves as usual.
- $A'_{lm}$  and  $B'_{lm}$  are determined by matching the value and derivative on the sphere boundary to zero. Thus this part  $(A'_{lm}u_l(\mathbf{r})+B'_{lm}\dot{u}_l(\mathbf{r})+u^{(2)}_l(\mathbf{r})) Y_{lm}(\mathbf{r})$  is formally a local orbital.

# THE LAPW+LO METHOD

## Key Points:

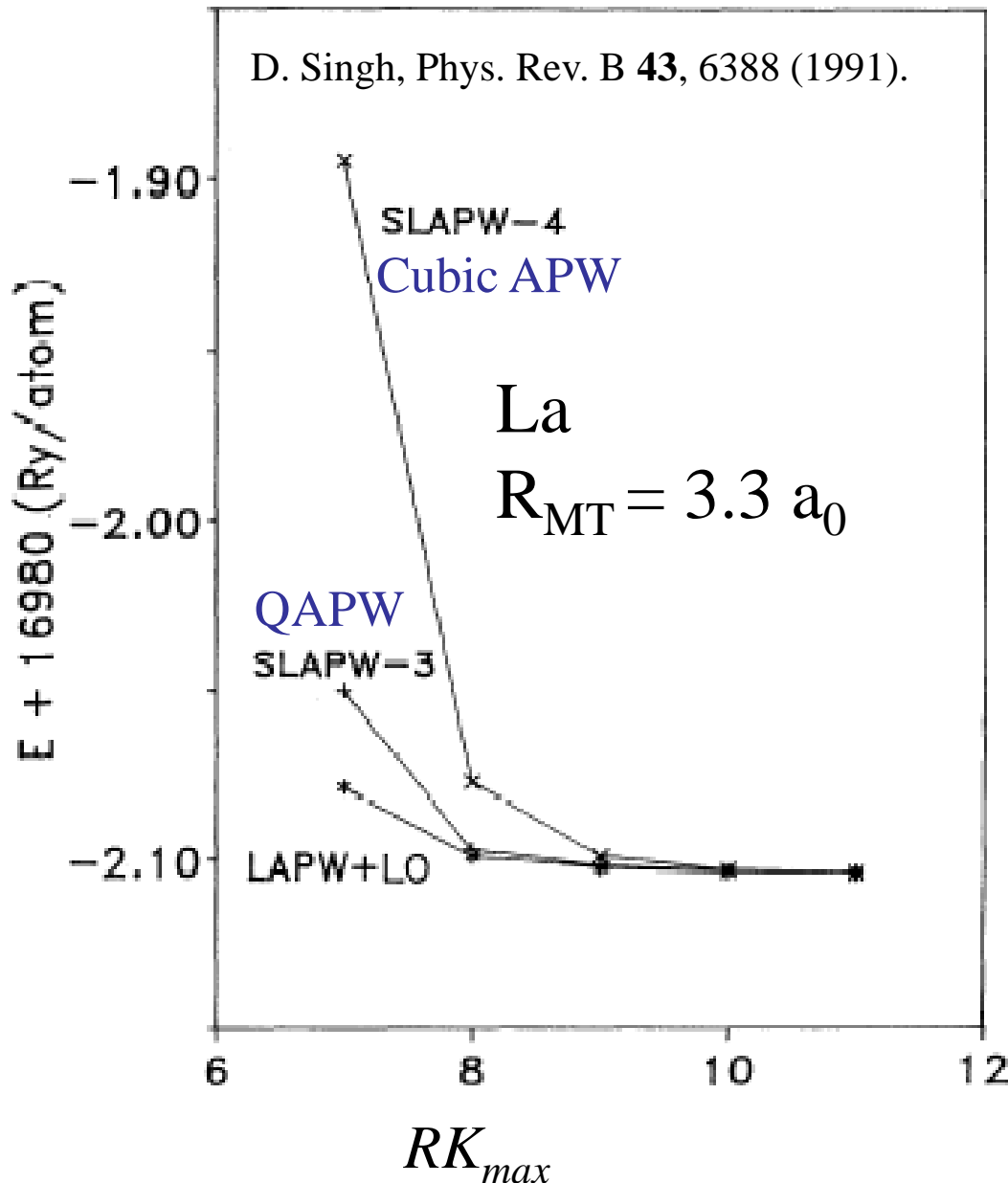
1. The local orbitals need (and should) only be used for those atoms and angular momenta where they are needed.
2. The local orbitals do not serve as surrogate atomic wavefunctions in the sense that they are in mixed basis plane wave codes: They are just another way to handle the augmentation. They look very different from atomic functions.
3. We are trading a large number of extra plane wave coefficients for some  $c_{lm}$ .

## Shape of H and S



# THE LAPW+LO METHOD

D. Singh, Phys. Rev. B **43**, 6388 (1991).



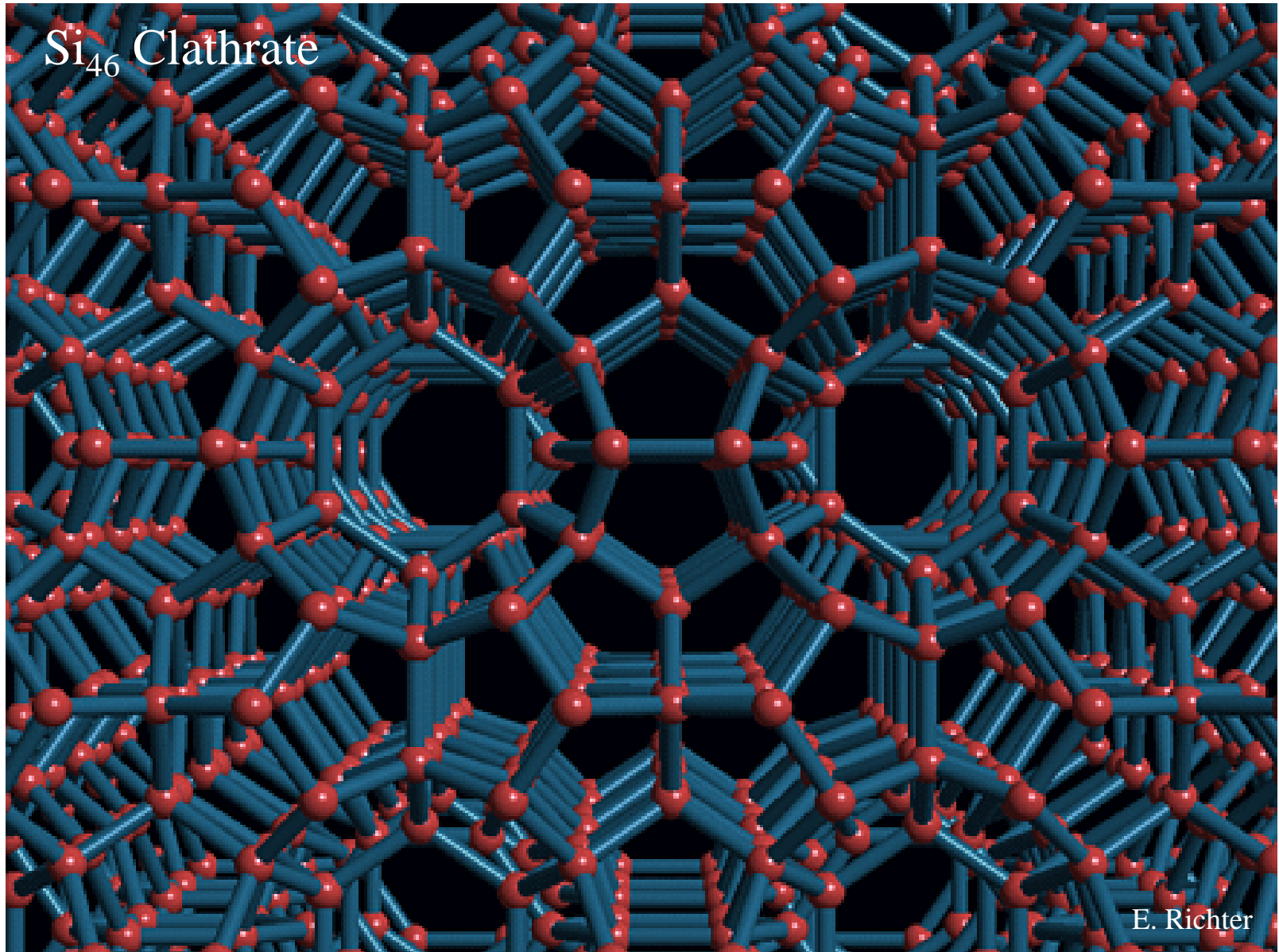
LAPW+LO converges like LAPW. The LO adds a few basis functions (i.e. 3 per atom for p states). Can also use LO to relax linearization errors, e.g. for a narrow  $d$  or  $f$  band.

Suggested settings:

Two “energy” parameters, one for  $u$  and  $\dot{u}$  and the other for  $u^{(2)}$ . Choose one at the semi-core position and the other at the valence.



# THE COST OF PLANEWAVES



Example of a structure with short bonds and large open spaces

# THE APW+LO METHOD

In certain cases it is highly advantageous to lower  $RK_{\text{MAX}}$  even at the expense of some local orbitals:

- Structures with short bonds and large empty spaces.
- Structures with some “hard” atoms embedded in a matrix of “soft” atoms: *e.g.* Mn impurities in Ge.

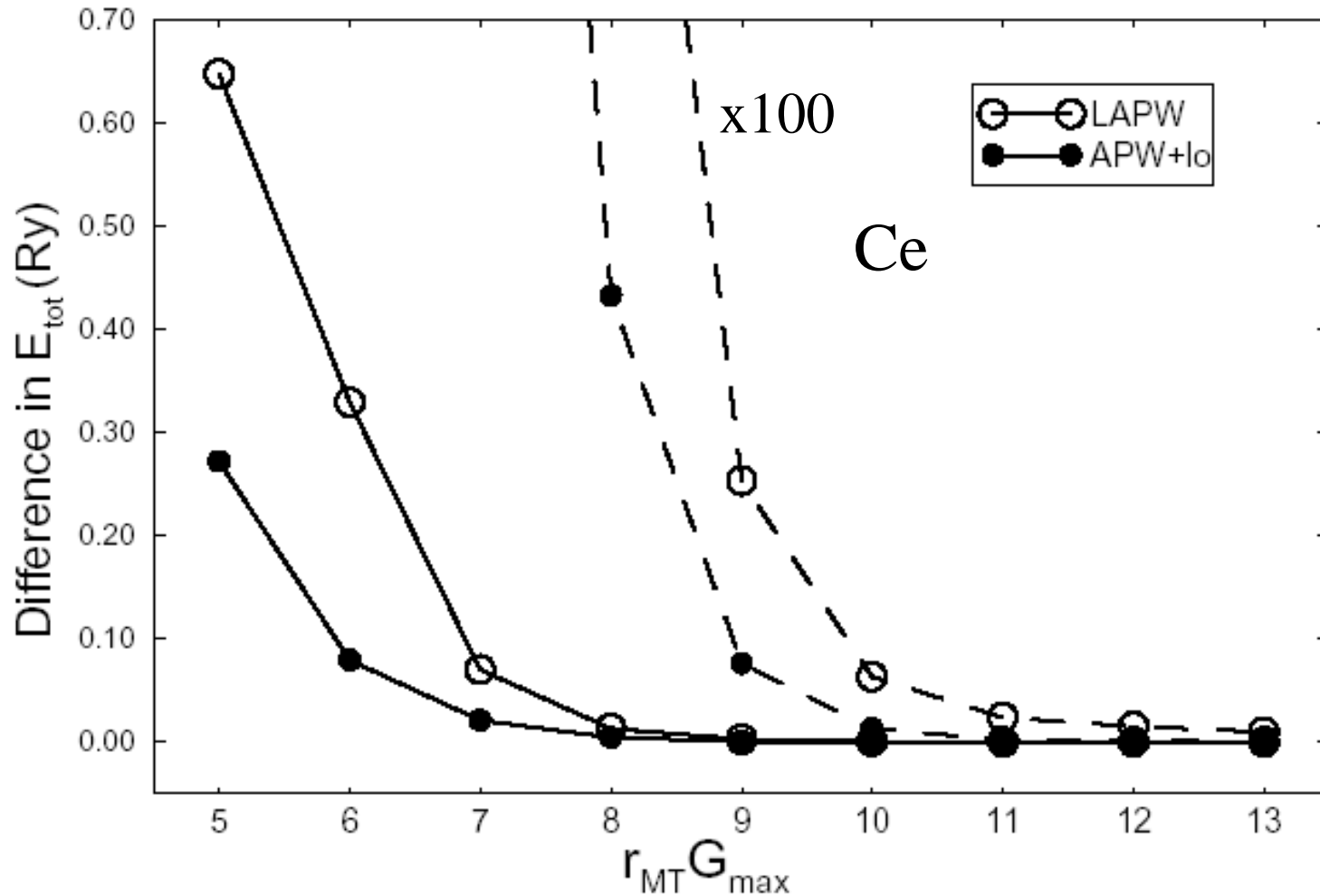
Then it is advantageous *for selected atoms and  $l$* , to use local orbitals to go back to the APW method.

$$\varphi(\mathbf{r}) = \left\{ \begin{array}{l} \Omega^{-1/2} \sum_{\mathbf{G}} c_{\mathbf{G}} e^{i(\mathbf{G}+\mathbf{k})\cdot\mathbf{r}} \\ \sum_{lm} (A_{lm} u_l(\mathbf{r})) Y_{lm}(\mathbf{r}) + \\ \sum_{lm} c_{lm} (A'_{lm} u_l(\mathbf{r}) + u^{(2)}_l(\mathbf{r})) Y_{lm}(\mathbf{r}) \end{array} \right.$$

*n.b.* now we only match the value on the boundary for these  $l$ . This means that there are extra APW-like kinetic energy terms in the Hamiltonian and forces.

# Convergence of the APW+LO Method

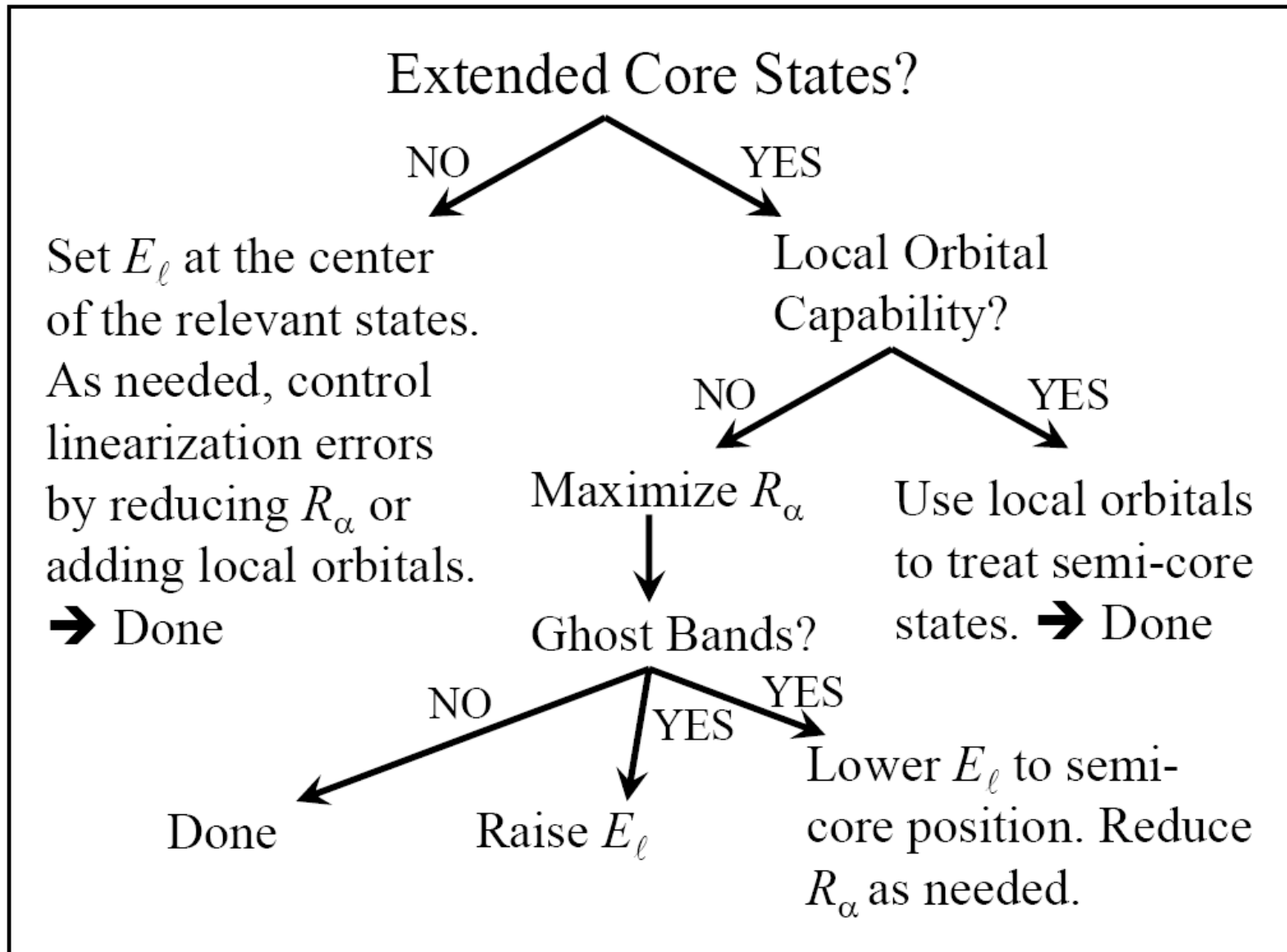
E. Sjöstedt, L. Nordström and D.J. Singh, Solid State Commun. **114**, 15 (2000).



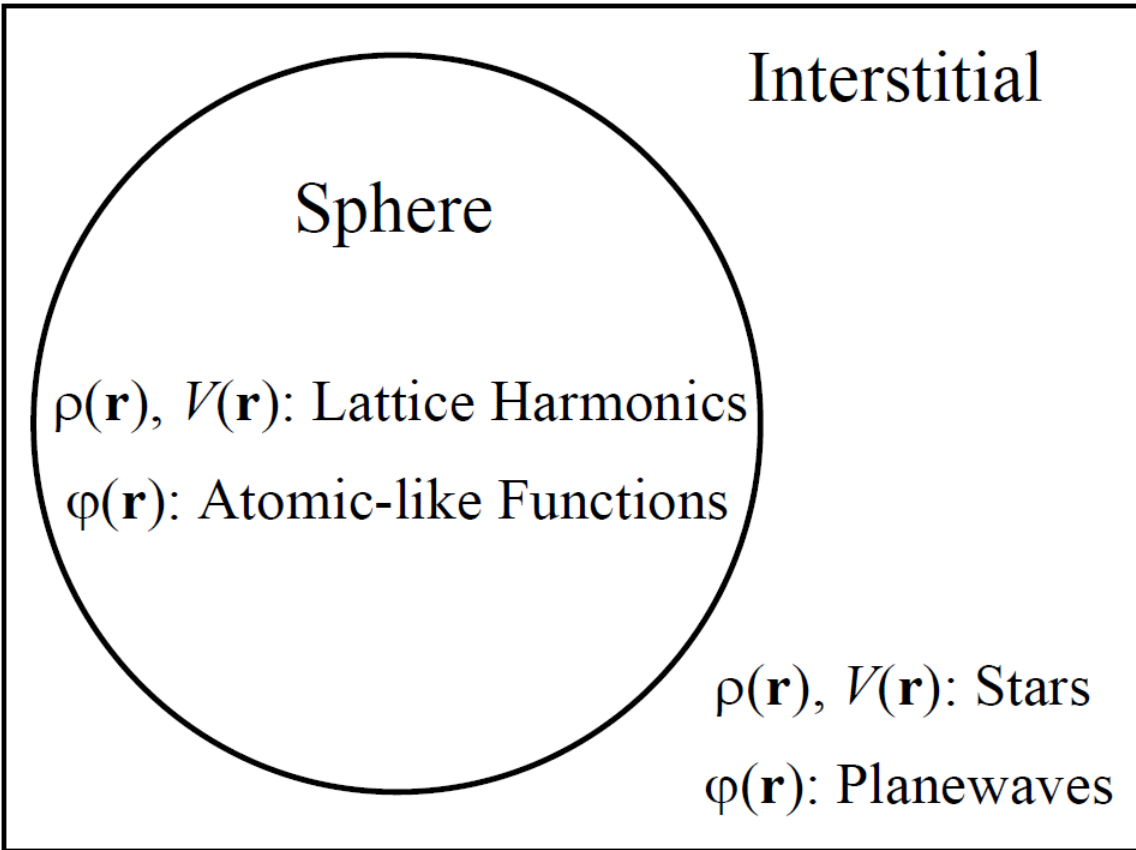
# REMARKS ON THE APW+LO METHOD

- APW+LO is equivalent to LAPW not LAPW+LO. It is not suitable for handling semicore states. For this LAPW+LO or APW+2LO should be used.
- There is no requirement that all atoms or angular momenta be augmented in the same way (see Madsen *et al.*). This can be exploited by using APW+LO *only for those atoms and  $l$  for which a high  $G_{max}$  would otherwise be needed*. For example, with Mn in Ge one might use APW+LO only for the Mn 3d channel, and LAPW for all others.

# How to Set Linearization Parameters



# Charge Density, Potential, etc.

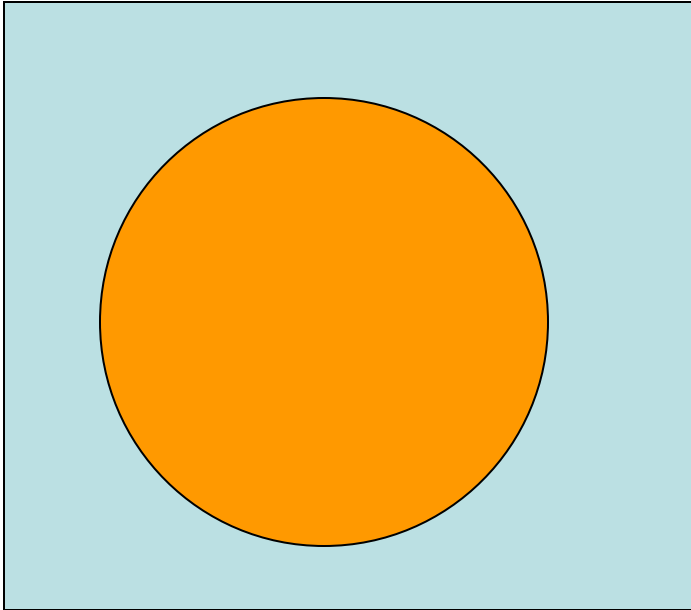


- Normally exploit lattice symmetry:
  - Stars in interstitial.
  - Lattice harmonics in spheres.
  - Only store for inequivalent atoms.

Allows for fast evaluation of Coulomb potential via multipole approach.

# Multipole Method for Coulomb Potential

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1. Evaluate multipole moments of  $\rho$  inside spheres.
2. Construct a smooth charge density (the pseudocharge) that is the same as the real charge outside the sphere, and has the same multipoles inside
  - n.b. can construct a smooth charge with a given multipole that is zero outside a sphere

3. Use Fourier transform method to get  $V \rightarrow$  exact in interstitial, but not in spheres.
4. Integrate Poisson's equation inward from sphere on radial grids to get  $V$  inside spheres.

$\rightarrow$  fast method comparable to planewaves

$$V_C(\mathbf{G}) = \frac{4\pi\rho(\mathbf{G})}{|\mathbf{G}|^2}$$

# References

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